



Efficient MgO-based mesoporous CO₂ trapper and its performance at high temperature

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ABSTRACT

A novel MgO-based porous adsorbent has been synthesized in a facile co-precipitation method for the first time, in order to provide a candidate for trapping CO₂ in flue gas at high temperature. The resulting composite exhibits a mesoporous structure with a wide pore size distribution, due to the even dispersion and distribution of microcrystalline MgO in the framework of alumina to form a concrete-like structure. These sorbents can capture CO₂ at high temperature (150–400 °C), possessing high reactivity and stability in cyclic adsorption–desorption processes, providing competitive candidates to control CO₂ emission.

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1. Introduction

Combustion of fossil fuels emits huge amounts of carbon dioxide, contributing to global climate changes [1,2]. Therefore it is necessary to trap CO₂ from flue gas at first, and then sequester CO₂ via geological formations, mineralization and ocean storage [3–8]. Various efficient methods have been proposed for CO₂ capture for this purpose, including commercial liquid amine, carbon-based adsorbent and amine-modified mesoporous materials [9–12]. However, most of these methods are carried out at relative low temperature, generally below 100 °C, while the temperature of flue gas vent is usually in the range of 150–400 °C [13,14]. Zeolites and hydrotalcite have been also tested for CO₂ adsorption [15,16], but they are flaccid at high temperatures. Thus, it is crucial to develop new functional material and technique to trap CO₂ from flue gas at high temperature.

Recently, adsorbents based on alkali metal oxide and alkali earth metal oxides with considerable CO₂ uptake and reversibility have been developed [17–19]. Among them Li-based adsorbents is able to trap CO₂; however, decomposition of Li₂CO₃ needs high temperature (800–900 °C) to release CO₂. Moreover, they are expensive [20–22]. Ca-based sorbents can capture CO₂ at 600–700 °C, but temperature above 900 °C is required for their regeneration [23,24].

The relatively high temperature range of the flue gas enables magnesia to be a candidate for CO₂ capture. MgO loaded on Al₂O₃ sorbents can be prepared by impregnation method, and their CO₂ adsorption capacities reach 42 and 60 mg g⁻¹ in the absence and presence of water vapor, respectively [25]. Other MgO-based sorbents are prepared by modification of dolomite for CO₂ removal in syngas at high temperature (300–450 °C) and high pressure (20 atm), and their capacity achieves 102–117 mg g⁻¹ [26]. Mesoporous silica SBA-15 is treated with sucrose and sulphuric acid to obtain mesoporous carbon (CMK-3) replica through the nano-casting process. With a similar procedure the mesoporous magnesia is then replicated from the mesoporous carbon CMK-3, and it exhibits a maximum CO₂ adsorption capacity of 100 mg g⁻¹ at 100 °C [27]. Nevertheless, its application is limited by the complicated preparation and the high cost. Thus, it is necessary to develop new CO₂ trappers with high capacity at high temperature and atmospheric pressure.

Herein, a new MgO-based mesoporous adsorbent is synthesized using a co-precipitation method, which allows both the synthesis and the modification of mesoporous alumina to be performed in one-pot process. As the result, a mass of MgO can be dispersed in the sorbent, resulting in a composite having considerably large surface and pore volume. The new adsorbents are used for CO₂ capture in the temperature range of 150–400 °C under atmospheric pressure. Since the flue gas usually contains 8–17 vol.% of water vapor in the practical condition [24], these MgO-based composites have been also assessed their actual performance of trapping CO₂ in the presence of vapor.

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2. Experimental

In the typical synthesis, 6.38 g P123, poly(ethylene oxide)–poly(propylene oxide)–poly(ethylene oxide) (EO–PO–EO) triblock co-polymers, were dissolved in 20 g H₂O, followed by the addition of Al(NO₃)₃·9H₂O (0.05 mole). The resulting mixture was stirred at 40 °C for 24 h to form a clear sol and then aged at 45 °C for 6 h. Thereafter, a solution containing different amount of Mg(NO₃)₂·6H₂O and 20 g H₂O was dropwise added under gentle stirring conditions. After the sol was further aged of 6 h, a stoichiometric amount of NH₃·H₂O containing 0.2–0.3 mol of OH[−] was slowly added under slow stirring, and then the gel was transferred into a teflonlined autoclave for hydrothermal treatment at 100 °C for 24 h. Afterward, it was transferred to a beaker for evaporation at 90 °C. Finally, the solid was calcined at 600 °C for 4 h, converting boehmite to γ -Al₂O₃ while the incorporated Mg(OH)₂ was transformed into MgO. The samples were named as 5AnM, where 5 and *n* represent the molar quantities of Al and Mg, respectively. For example, 5A5M composite was prepared with 0.05 mol aluminum and 0.05 mol magnesium, while 5A0M meant the absence of magnesium. The mass fraction of MgO incorporated into the sample of 5A3M, 5A5M and 5A7M is 24%, 39% and 55%, respectively. For comparison, MgO was obtained from the calcination of Mg(NO₃)₂·6H₂O at 600 °C.

X-ray diffraction (XRD) patterns of sample were recorded using an ARL XTRA diffractometer (power 40 kV, 40 mA) with Cu-K α radiation in the 2-theta range from 0.5° to 8° or from 6° to 80°. N₂ adsorption–desorption isotherms were measured on a Micromeritics ASAP 2020 system at −196 °C, being the sample outgassed at 300 °C for 4 h prior to test. The Brunauer–Emmett–Teller (BET) method was utilized to calculate the specific surface areas of sample. The pore size distribution was derived from the adsorption branch of the isotherm using the Barrett–Joyner–Halanda (BJH) algorithm, and the total pore volume was estimated from the amount adsorbed at a relative pressure (*P/P*₀) of 0.99. SEM images and X-ray mapping images of sample were obtained with Hitachi S4800 microscopes at 20 kV, 10 mA and 40 kV, 15 mA, respectively.

To perform the FTIR test of CO₂ adsorbed on sorbent, the self-supporting wafer with the density of 13 mg cm^{−2} was activated at 500 °C for 4 h under vacuum. Afterward the sample was taken the background spectrum at 25 °C, and then contacted with CO₂ at 200 °C for 0.5 h followed by evacuation of 0.5 h, prior to taking the spectrum at 25 °C. Then, the sample was kept with CO₂ at 300 °C, 400 °C, 500 °C for 0.5 h and then recorded at 25 °C by a VERTEX 70 FTIR spectrometer with a resolution of 2 cm^{−1}.

CO₂ adsorption was performed as previously reported [24]. In a typical process, 75 mg of granular sample with a size of 20–40 mesh were heated in a flow of N₂ (dry, 99.995%) at 600 °C for 2 h,

then cooled to the given temperature. The simulated flue gas containing CO₂, N₂ with or without H₂O was introduced to contact with the sample for 1 h, and its speed was kept 50 ml min^{−1} (room temperature). Afterward, the sample was purged with a N₂ flow for 10 min followed by the cooling down to room temperature. Temperature programmed desorption (TPD) of CO₂ was performed from room temperature to 600 °C at a rate of 8 °C min^{−1} in the flow of N₂, while the liberated CO₂ was detected by an “online” Varian 3380 gas chromatograph and quantitatively measured by the external standard method. A drying tube of P₂O₅ was used in the reactor outlet to drastically absorb the moisture desorbed from the sample, ensuring the CO₂ to be the sole entering gas into chromatograph. In the cyclic adsorption–desorption tests, the sample was allowed to re-adsorb CO₂ in the same manner aforementioned as soon as the entire desorption process was finished. TG-MS analysis of sample was conducted on a thermobalance (STA-499C, Netzsch) coupled with a mass analyzer (QUADSTAR-422, Pfeiffer). The composite adsorbed CO₂ in the reactor with the feed gas of 10 vol.% CO₂, 10 vol.% H₂O, and 80 vol.% N₂ at 200 °C in the manner aforementioned, and then was transferred for TG-DSC measurement. The CO₂ adsorbed sample was heated from room temperature to 600 °C at a rate of 8 °C min^{−1} and held at 600 °C for 1 h in a continuous flow of Ar (30 mL min^{−1}), and the effluent of desorption was simultaneously analyzed with mass spectrometer.

3. Results and discussion

3.1. Structural characterization of 5AnM composites

Fig. 1A displays low-angle XRD patterns of 5AnM samples. One diffraction with 2-theta value of about 1.1° is observed on the pattern, representing the mesoporous structure with no long-range order. As the Mg-content of sample increases, this diffraction peak lowers but not shift, indicating the declined ordered structure. The parent sample 5A0M exhibits typical diffraction lines of γ -Al₂O₃ (JCPDS 10-0425) with 2-theta values of 39.47°, 45.84° and 67.00° (Fig. 1B). Other 5AnM composites possess the diffraction peaks of MgO at 2 θ of 42.99° and 62.34°, and these peaks become slightly stronger and broader as the amount of incorporated Mg species increases. The (400) and (440) diffraction lines of alumina shift to lower diffraction angles after introduction of magnesia, because they are overlapped with the diffraction lines of the guest. As the result, three main diffraction peaks on the pattern of 5A7M can be attributed to that of magnesia (Fig. 1B). Although a lot of MgO is incorporated into the composite such as 5A7M, no sharp and dramatically increased XRD peaks emerge on the pattern, because microcrystalline structure of MgO forms in the framework [29].

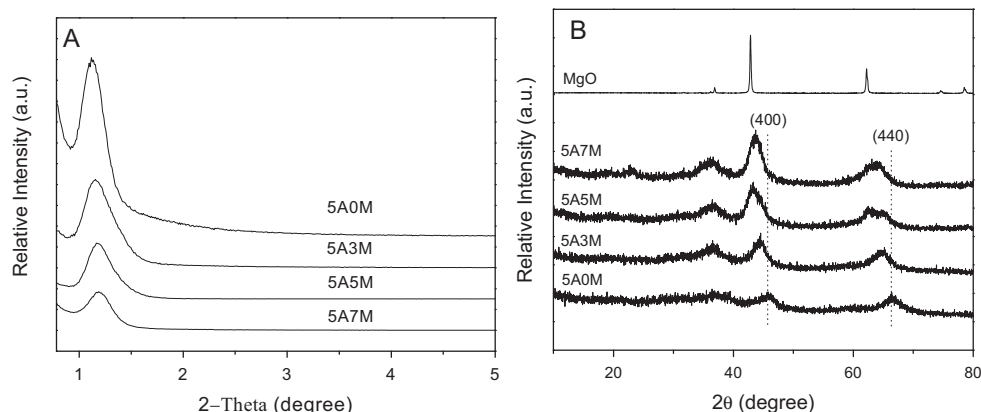


Fig. 1. Low-angle (A) and wide-angle (B) XRD patterns of 5AnM sample and MgO.

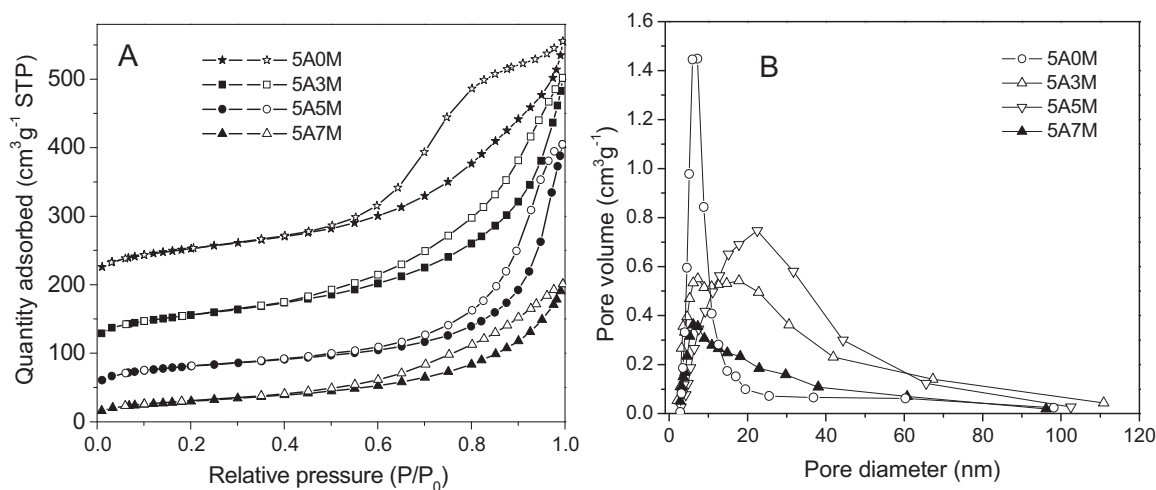


Fig. 2. Nitrogen sorption isotherms (A) and pore size distributions of 5AnM samples (B). In nitrogen sorption isotherms (A), closed and open symbols are used to represent the adsorption and desorption isotherms, respectively.

Fig. 2 depicts the N_2 adsorption–desorption isotherms and pore size distributions of 5AnM samples. All isotherms are of classical type IV with an obvious hysteresis loop, indicating the mesoporosity of these composites. N_2 uptake emerges at relative pressure of 0.5–0.99 in the isotherms of 5AnM samples (Fig. 2A), indicating the formation of mesopore with wide pore sizes (4–80 nm). The hysteresis loops become smaller with the increased Mg-content of samples, implying the declined mesoporous structure. At the same time, the surface area and pore volume are decreased as reported in Table 1. Parent 5A0M has a relatively small average pore diameter of 7 nm, while other 5AnM samples have average pore diameter of about 10 nm due to incorporation of magnesia. Mg-containing samples show significant variation of pore size as 5A3M has two sizes (Fig. 2B). These results show that the composite of 5AnM has a mesoporous structure with wide pore size distribution. Removing P123 surfactant template and converting boehmite to crystalline γ - Al_2O_3 in calcination procedure can generate the mesostructure with small pores [30]. The introduced Mg species seem to be incorporated into the framework of alumina, reducing the order of framework and lowering pore volume and surface area, while widening the average pore size of the material.

Fig. 3A shows the SEM image of 5A5M sample. It composes of irregular particles with some cumulate pores as shown in the inserted image with high magnification. Figs. 3B and S1 depict the EDX spectrum of 5A5M, 5A3M and 5A7M samples, respectively, showing the presence of Mg and Al. The Mg/Al atomic ratio of the samples is reported in Table 1. Even though several random areas are selected for EDX analysis, the detected Mg/Al molar ratio of 5A3M, 5A5M and 5A7M is 0.64, 0.95 and 1.42, respectively, almost same as that calculated (Table 1). This result suggests a homogeneous distribution of Mg species in these composites. Figs. 3C, D and

S2 show the X-ray mapping of 5A3M, 5A5M and 5A7M, in which the microcrystalline MgO and Al_2O_3 seem to be even dispersed in nano level.

Fig. 4 reports the IR spectra of CO_2 adsorbed on 5A5M sample at 200 °C, in which the O–C–O stretching vibration at 1320 cm^{-1} can be attributed to bidentate carbonate and that at 1511 and 1590 cm^{-1} ascribed to unidentate carbonate species [31]. As the sample is evacuated at higher temperature, these peaks gradually decline. Finally bidentate carbonate species disappear after the evacuation at 500 °C while unidentate ones survive but their intensity is dramatically lowered.

3.2. CO_2 trapping by MgO-based porous composite

Table 1 reports the CO_2 adsorption on 5AnM samples detected by TPD method. Both parent 5A0M and MgO itself have a weak adsorption capacity of 6 and 9 mg g^{-1} at 200 °C. Incorporation of magnesia allows 5AnM samples to trap CO_2 amounts several times higher than their parent in the same conditions. In particular, 5A3M, 5A5M and 5A7M samples adsorbed 45, 77 and 37 mg of CO_2 per gram, respectively. Two factors should be taken into account for the phenomenon: the homogeneous dispersion of MgO in Al_2O_3 framework and the mesopores in the composite, which ensures CO_2 to easily contact with the active sites. Different situation emerges if the adsorption efficiency of sample is calculated by its surface area. MgO is the champion and 5A5M becomes the runner-up. Either 5A3M or 5A7M show the lower efficiency than 5A5M, revealing the property–function relation of magnesia guest. The CO_2 adsorption capacity of sample relates to the amount of adsorption site available [12], which may be a balance of the content of Mg species and their dispersion in the

Table 1
The textural properties and CO_2 adsorption capacities of 5AnM samples.

Sample	MgO	5A0M	5A3M	5A5M	5A7M
BET surface area ($\text{m}^2\text{ g}^{-1}$)	About 10	253	202	177	108
Pore volume ($\text{cm}^3\text{ g}^{-1}$)	–	0.52	0.48	0.45	0.26
D_p (nm)	–	7.2	10.2	10.3	9.8
Mg/Al atomic ratio calculated	–	0	0.6	1.0	1.4
Detected by EDX method	–	0	0.64	0.95	1.42
CO_2 trapped at 200 °C (mg g^{-1})	9	6	45	77	37
(mg m^{-2} , A)	0.9	0.02	0.22	0.43	0.34
CO_2 trapped at 200 °C with vapor (mg g^{-1})	37	7	93	131	84
(mg m^{-2} , B)	3.7	0.03	0.46	0.74	0.78
B/A	4.1	–	2.09	1.72	2.29

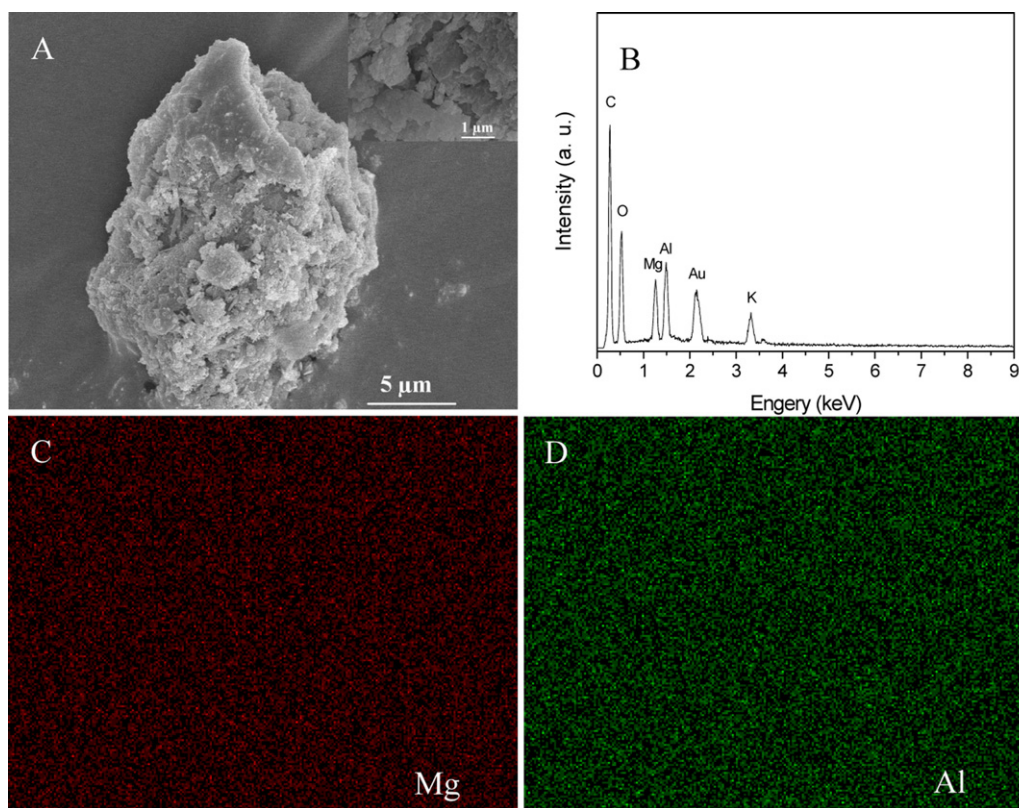


Fig. 3. SEM images (A), EDX spectrum (B), and X-ray mapping (C, D) of 5A5M sample, in which the magnification is 5,00,000.

surface of samples. Since 5A5M composite exhibits the highest adsorption performance, it is selected for further investigations.

Fig. 5A displays the TPD profile of CO₂ from 5A5M samples in the absence or presence of water vapor. Two desorption peaks are observed at 211 °C and 463 °C in the case of adsorbing CO₂ at 200 °C, and they can be tentatively assigned to two kinds of adsorption sites at 200 °C [25]. Once the water vapor is introduced in adsorption procedure, the climax of desorption peaks shift to higher temperature, 333 °C and 524 °C, while the amount of desorption is obviously increased. Similar phenomena also emerge on 5A3M and 5A7M, but absent on parent 5A0M sample (Table 1). The amount of CO₂ desorbed from 5A3M, 5A5M and 5A7M is enlarged about 110%, 70% and 130% (Figs. 5B and S3), respectively. However, almost all of the trapped CO₂ can be desorbed from the sample at 600 °C (Fig. 5A), which will facilitate the regeneration and cycle utilization of the CO₂-trapper. The function of vapor to enhance the capability of MgO-based composite in trapping CO₂ at 200 °C may involve the mechanism. The main reaction between CO₂ and MgO at high temperature without vapor is the formation of carbonate as described: MgO + CO₂ = MgCO₃ [29]. With the introduction of water vapor, a transient Mg(OH)₂ component is distinctly formed when the MgO is exposed to the steam environment [32], and then the capture process of CO₂ involves the reactions: MgO + H₂O = Mg(OH)₂, Mg(OH)₂ + CO₂ = MgCO₃ + H₂O. It is suggested that the kinetics of CO₂ capture by Mg(OH)₂ is much faster than MgO [33]. Therefore, 5AnM adsorbents can capture more CO₂ from feed gas. Apparently, this feature of 5AnM composites is beneficial for their potential application for capturing CO₂ under practical conditions, because flue gas contains water vapor [24].

Fig. 6 illustrates the TG-DSC and MS analysis on the CO₂ desorbed from 5A5M sample. The sample losses about 2% of weight at 300 °C, and an endothermic peak appears on the DSC spectrum. Upon heating the sample from 300 °C to 600 °C, the TG curve dramatically declines to indicate the further weight loss of about 10%

meanwhile a high endothermic peak emerges (Fig. 6A), mirroring the occurrence of desorption in this temperature range. CO₂ desorption is also confirmed by MS analysis (Fig. 6B), in which a large desorption of CO₂ is detected from 200 °C to 600 °C.

Fig. 7A examines the effect of adsorption temperature on the amount of CO₂ captured by 5A5M. The interaction of CO₂ with the MgO-based sample in the presence of vapor is an exothermic process, according to the DSC curve in Fig. 6A. And referring to literature [33] the reaction is MgO (s) + H₂O (g) = Mg(OH)₂ (s) ΔH = -81.48 kJ mol⁻¹, Mg(OH)₂ (s) + CO₂ (g) = MgCO₃ (s) + H₂O (g), ΔH = -19.66 kJ mol⁻¹, therefore this composite should exhibit

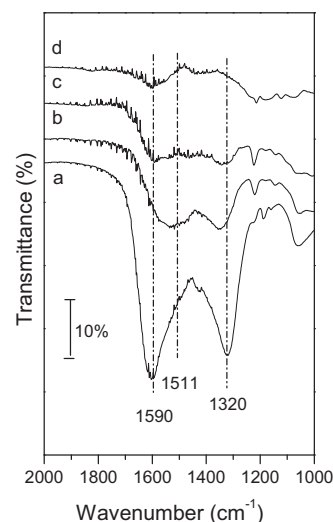


Fig. 4. FTIR difference spectra of CO₂ adsorbed on 5A5M sample at (a) 200 °C, (b) 300 °C, (c) 400 °C, and (d) 500 °C.

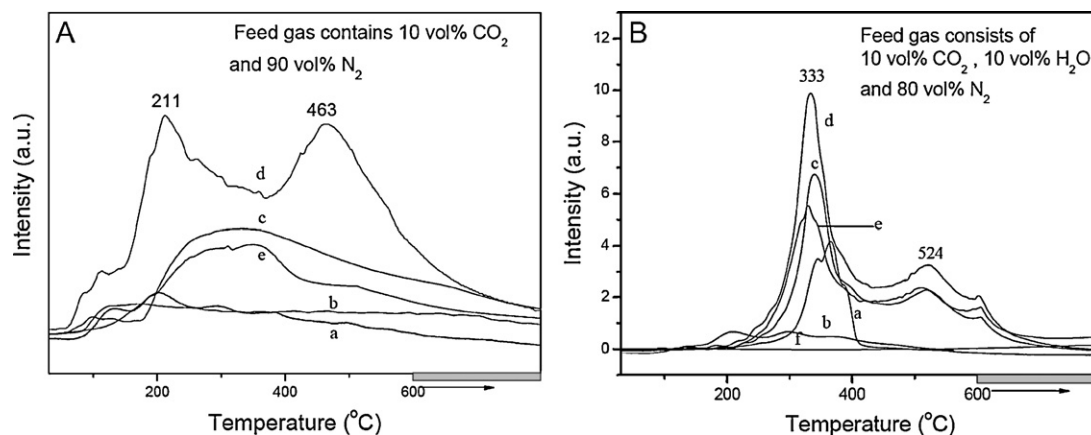


Fig. 5. CO₂-TPD profiles of samples in the absence (A) or presence (B) of vapor, (a) MgO, (b) 5A0M, (c) 5A3M, (d) 5A5M, (e) 5A7M, and (f) blank curve.

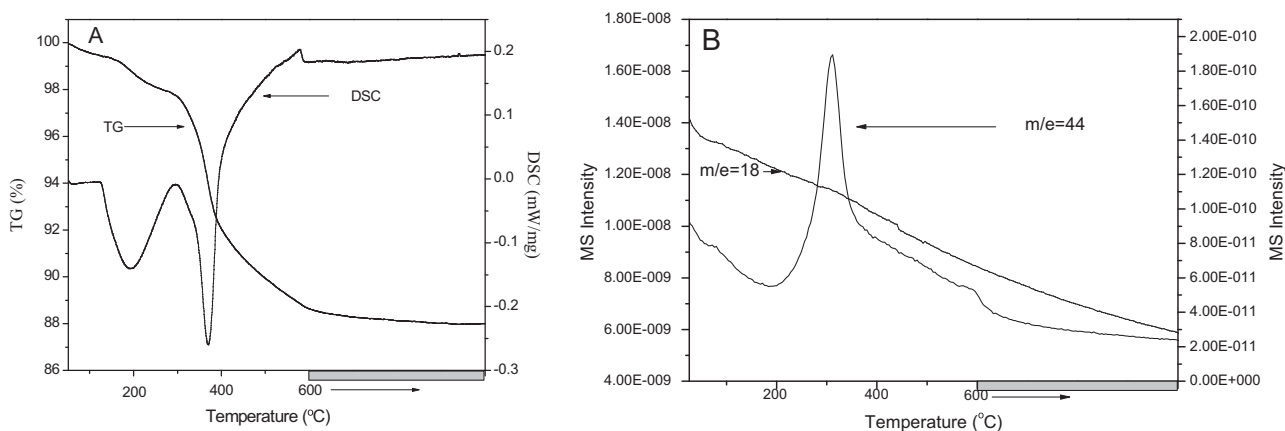


Fig. 6. TG-DSC (A) and MS (B) curves of CO₂ desorption from 5A5M sample.

a higher activity at relatively low temperature. However, the CO₂ trapping capacity by 5A5M sample is 117 mg g⁻¹ at 150 °C, lower than that at 200 °C (131 mg g⁻¹). At higher temperature the adsorption capacity decreases to 100 mg g⁻¹ at 300 °C and 41 mg g⁻¹ at 400 °C (Figs. 7A and S3). The interaction of CO₂ with 5A5M sample may be kinetically controlled at relatively low temperature hence the reactivity of the composite is slightly enhanced at 200 °C in comparison with that at 150 °C. Nonetheless, the rate of reverse reaction is increased at a higher temperature so that the

thermodynamics becomes dominant, leading to lower CO₂ adsorption capacities. Fig. 7B shows the cyclical capacity of 5A5M in the trapping-desorption of CO₂ at 200 °C with the presence of vapor. This sample exhibits a high CO₂ trapping capacity of 115 mg g⁻¹ in the 6th cycle, equaling to 88% of that in 1st cycle (131 mg g⁻¹). For practical use, the CO₂ trapper should not only possess a high capability, but also display a stable cyclic utilization performance. The high activity and stability of 5A5M composite in the trapping of CO₂ at 200 °C are beneficial for its potential practical applications.

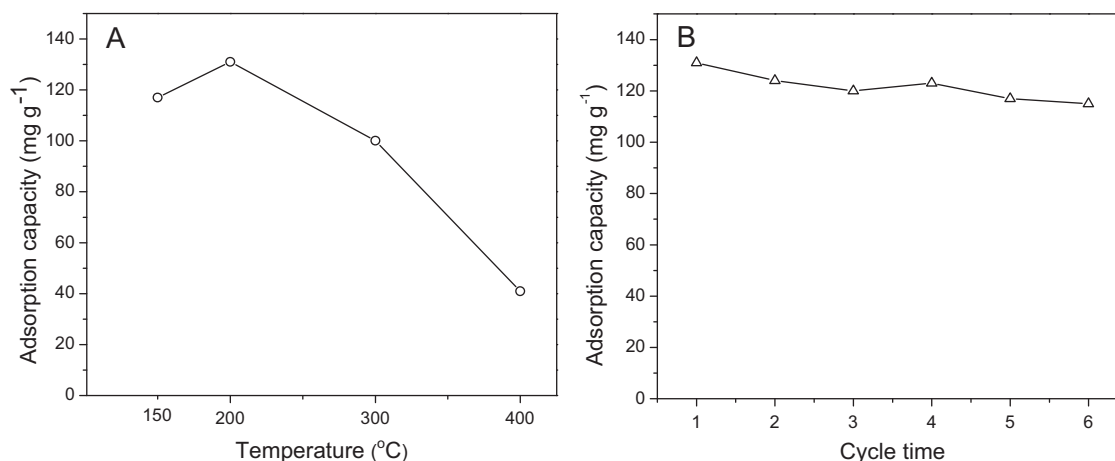


Fig. 7. CO₂-trapping (A) and the cycle utilization at 200 °C (B) of 5A5M sample.

4. Discussion

Fabrication of favorable structure/texture is crucial for the MgO based mesoporous composite to trap CO₂ at 200 °C. The capturer should have enough active sites with optimal accessibility to contact with CO₂, and then desorb CO₂ at a higher temperature. Therefore, the candidate needs to have a pore structure with a good flexibility to bear the structural variation in adsorption/reaction and desorption/regeneration. Moreover, the active component magnesia should be highly dispersed and felicitously separated in such porous structure in order to tolerate the cycle of formation and decomposition of carbonate.

Alumina and magnesia are chosen to fabricate the CO₂ capturer. Alumina fails to trap CO₂ at 200 °C (Table 1), but it can form the necessary mesoporous carrier, unlike silica that reacts with magnesium at high temperature to form the weak basic compounds [34]. Magnesia cannot assemble to mesoporous adsorbent in common hydrothermal synthesis, and MgO itself exhibits a poor capacity of trapping CO₂ (9 mg g⁻¹, Table 1) due to its small surface area. Consequently, we plan to disperse magnesia on the support of alumina in order to form desirable composite.

In our synthesis, aluminate reactant was allowed to contact with surfactant at first, in order to let Al species interact with P123 micelles through the hydrogen bonding between the surfactant and the nitrate anions [34]. And then magnesium salt incorporates inside the aggregation of host through co-precipitation. During the evaporation step in the synthesis cations may undergo complexation with the hydrophilic EO groups in P123 upon the evaporation of water [34], and these interactions among PEO-Mn⁺-anion promote the dispersion level of guest oxide species in the carrier, probably through insertion and/or micro-crystallization. Judged from the XRD pattern of 5AnM samples (Fig. 1), formation of hydrotalcite is excluded, and the MgO is not simply coated on the mesoporous alumina. Otherwise, as the Mg-content of sample increased from 24% (5A3M) to 55% (5A7M), the XRD pattern of MgO phase would be drastically strengthened, as that previously reported [28,35]. Rather, based on the results of XRD and X-ray mapping, it seems that magnesia has been inserted into the framework of alumina, probably through micro-crystalline form, resembling that of concrete, in which the magnesia micro-crystals equably insert into the framework formed by alumina. Due to the even insertion of guest into the host framework, there is no surface enriching of MgO but the unique distribution of Mg and Al (Table 1). The surface Al/Mg molar ratio of composite determined by X-ray mapping method is close to that calculated. However, this insertion affects the pore structure of resulting composite, the pore size distribution is widened and the most probably pore size is enlarged (Fig. 2). As the Mg-content of composite increases, its pore structure is distorted and it seems to be non-ordered. Fortunately, it still owns a considerably high capability of capture CO₂ and the stability in recycle use at 200 °C (Figs. 5 and 7), thanks to the high dispersion of magnesia in alumina support.

Magnesia guests are accumulated to be micro-crystals in the composite, and only the ones on the top contact with CO₂. Thus, the property–function relation of magnesia in 5AnM composites is assessed by the efficiency of per square meter surface area. Mesoporous alumina has a very weak ability, and this data increases as magnesia is incorporated (Table 1), achieving the maximum at 5A5M sample (0.43 mg m⁻²). Considering the surface consisting of equal molar magnesia and alumina, this value is close to that of MgO itself in the adsorption of CO₂ at 200 °C in dry gas stream: since the magnesia was diluted with equal mole of alumina in this composite where equal molar alumina exist in MgO with an even dispersion, and alumina has a very low adsorption efficiency of CO₂ (0.02 mg m⁻²), the mean adsorption value of MgO in the composite should divide 2 to be equal to 0.45 mg m⁻² (0.9/2 = 0.45 mg m⁻²).

Presence of vapor promotes the activity of 5AnM composite in trapping of CO₂ at 200 °C (Figs. 5 and S3), through the formation of Mg(OH)₂ that can fast react with CO₂ [32]. Moreover, the difference among the efficiency of three 5AnM samples (Table 1) is decreased, among them 5A7M exhibits the highest promotion. On the other hand, the efficiency of MgO in trapping CO₂ at 200 °C with vapor is much higher than that of 5A5M (about 5 times, Table 1), which implies the different property–function relation between them. If we can optimize the distribution of magnesia guest in porous support, it is hopeful to enhance the performance of trapping CO₂ at high temperature.

5. Conclusion

Some conclusive remarks can be tentatively derived from the results mentioned above.

- (1) A new MgO-based mesoporous composite has been successfully synthesized by an one-pot synthesis strategy.
- (2) With the microcrystalline MgO existed in the framework of alumina, this composite can capture amounts of CO₂ of 77 and 131 mg g⁻¹ at 200 °C in the absence or presence of water vapor, respectively.
- (3) This CO₂-trapper can be regenerable at 600 °C, being stable for cyclic adsorption.

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Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at doi:10.1016/j.jhazmat.2011.12.036.

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